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ABSTRACT
Using density functional theory together with dispersion correction, we optimize geometries and calculate encapsulation energies, charge transfer, and electronic structures of a one-dimensional \( \text{ThCl}_6 \) chain encapsulated within single-walled carbon nanotubes. The optimized structures are in agreement with the experimentally observed single chain of \( \text{ThCl}_6 \) confined within nanotubes. Exoergic encapsulation energies indicate that the chain is thermodynamically stable inside the nanotubes. The inclusion of dispersion correction enhanced the encapsulation. The strong nature of encapsulation is further confirmed by the charge transfer between the nanotubes and the chain. The one-dimensional chain structure exhibits a small increase (0.20 eV) in the bandgap compared to that of bulk \( \text{ThCl}_4 \). The metallic (15,0) tube becomes a narrow gap semiconductor upon encapsulation, while the semiconducting nature of the (17,0) tube is still kept despite the reduction in the bandgap.

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I. INTRODUCTION
Single-walled nanotubes (SWNTs) are of great interest for their ability to encapsulate a variety of one-dimensional (1D) inorganic nanocrystals.\(^1\)\(^-\)\(^6\) The synthesis of SWNTs filled with inorganic nanocrystals has been considered as a possible route for studying the properties of low-dimensional materials and tuning the electronic properties of nanotubes.\(^7\)\(^-\)\(^9\) Much experimental work has been reported on the synthesis and characterization of metal halides and metal chalcogenides encapsulated within SWNTs.\(^4\)\(^,\)\(^5\)\(^,\)\(^7\)\(^-\)\(^11\)\(^-\)\(^14\) In general, encapsulation resulted in a change in the encapsulated material relative to its bulk. A high resolution transmission electron microscopy (HRTEM) study on the filling of KI resulted in the formation of 1D 2\( \times \)2 KI crystals in which both K and I form a 4:4 coordination in contrast to the 6:6 coordination observed in the bulk KI.\(^1\) A similar low-dimensional structure with 4:4 coordination has also been observed in the experimental study of filling SWNTs with SnSe.\(^15\) A unique 1D tubular structure of HgTe inside SWNTs was characterized by Carter et al.\(^1\) upon the filling of molten HgTe. In this tubular structure, both Hg and Te form three-dimensional coordination in contrast to the tetrahedral coordination observed for both He and Te in the bulk HgTe zinc blende structure.

In another experimental study, Sloan et al.\(^2\) filled SWNTs with ThCl\(_4\) and recorded their images using HRTEM. The encapsulated structure is observed to be a one-dimensional chain consisting of \( \text{ThCl}_6 \) polyhedral units along the tube axis and the diameter of the tube was reported to be 1.1 nm.\(^2\) Furthermore, the polyhedral units are eight-coordinate and share their edges as observed in the bulk ThCl\(_4\). The encapsulation of ThCl\(_4\) within SWNTs can be an ideal route to store Th as SWNT walls can provide protection from the chemical environment and external interactions.

Computational modeling techniques have been supportive in validating the experimentally observed structures within SWNTs and predicting the structural and electronic properties of the resultant encapsulated structures.\(^16\)\(^-\)\(^18\) Significant effort has been devoted to study one-dimensional nanocrystals encapsulated within SWNTs theoretically.\(^19\)\(^-\)\(^21\) Density functional theory (DFT)
simulations have been performed to validate the structures of 1D inorganic crystals, such as KI, HgTe, PbI$_2$, and SnSe, encapsulated within SWNTs and study the electronic properties of the composites formed. In some simulation work, model structures have been proposed for crystals such as CuF$_2$ and GeTe as lighter atoms are generally challenging to elucidate from HRTEM experiments.

In this study, we use spin-polarized DFT simulations to validate the experimentally observed 1D ThCl$_6$ chain confined within SWNTs and determine the nature of the interaction between the tube and chain. Furthermore, simulations enabled us to calculate the charge transferred between the guest and the host and electronic nature of the resultant composite relative to that of pristine SWNT.

II. COMPUTATIONAL METHODS

All calculations were performed using spin-polarized mode of DFT as implemented in the VASP (Vienna Ab initio simulation package) code. The exchange–correlation term was modeled using the generalized gradient approximation (GGA) parameterized by Perdew, Burke, and Ernzerhof (PBE). The valence electronic configurations for C, Th, and Cl were $2s^2 2p^2$, $6s^2 6p^6 6d^2 7s^2$, and $3s^2 3p^5$, respectively. The standard projected augmented wave (PAW) potentials and a plane-wave basis set with a cutoff of 500 eV were employed. We used an $8 \times 8 \times 8$ Monkhorst–Pack $k$-point mesh for the bulk ThCl$_6$ structure. For the isolated 1D ThCl$_6$ chain structure and its structure encapsulated within SWNTs, a $4 \times 1 \times 1$ Monkhorst–Pack $k$-point mesh was used. Structure optimizations were carried out using a conjugate gradient algorithm, and the Hellman–Feynman theorem with Pulay corrections was used to obtain the forces on the atoms. In all relaxed structures, forces on the atoms were smaller than 0.01 eV/Å.

For calculations on the infinite 1D ThCl$_6$ chain and ThCl$_6$@SWNTs, periodic boundary conditions were applied to enforce a minimum lateral separation of 25 Å between structures in adjacent unit cells. The encapsulation energy of the ThCl$_6$ chain can be calculated by considering the difference in the total energy of the ThCl$_6$@SWNT and the total energies calculated for an isolated ThCl$_6$ chain structure and an isolated ThCl$_6$,

$$E_{enc} = E_{ThCl6@SWNT} - E_{SWNT} - E_{ThCl6},$$

where $E_{ThCl6@SWNT}$ is the total energy of ThCl$_6$ encapsulated within a SWNT and $E_{SWNT}$ and $E_{ThCl6}$ are the total energies of a SWNT and an isolated gas phase ThCl$_6$ chain, respectively.

Initial configurations, optimized geometries, and charge density plots were visualized using a 3D visualization program VESTA (Visualization for Electronic and Structural Analysis). Density of state (DOS) plots were visualized and analyzed using XMGRACE software. Bader charge analysis enabled us to calculate the charge transferred between the nanotubes and the 1D ThCl$_6$ crystal. In this method, electronic charges on individual atoms in the relaxed configurations are calculated based on the partitioning method as implemented by Bader.

III. RESULTS AND DISCUSSION

A. Crystal structure of ThCl$_6$

ThCl$_6$ crystallizes in the tetragonal $I4_1/amd$ space group. Its experimental lattice parameters measured at room temperature are reported to be $a = b = 8.486$, $c = 7.465$ Å, and $a = β = γ = 90°$. Figure 1(a) shows its crystal structure in which each Th atom forms eightfold coordination with adjacent eight Cl atoms. The dodecahedron units formed by each Th are connected by their edges. There are two sets of Th–Cl bond distances [2.718 (4×) and 2.903 (4×)] present in each dodecahedron unit. In order to assess the quality of the pseudopotentials and basis sets used for C, Th, and Cl, we performed the full geometry optimization calculation on the crystal structure of ThCl$_6$ by relaxing positions of both atoms and the simulation box simultaneously. There is excellent agreement between the calculated and experimental values showing the efficacy of the simulation parameters in this study (see Table I).

B. 1D ThCl$_6$ chain structure

Next, we considered the optimization of the 1D chain structure of ThCl$_6$ with periodic boundary conditions in the absence of a confining tube. This chain structure consisted of four ThCl$_6$ units ($Th_4Cl_{12}$) in the supercell with dimensions of $16.97 \times 25 \times 25$ Å$^3$. The relaxed structure is depicted in Fig. 2(a). Table II reports the calculated bond distances, bond angles, and the bandgap. The optimized structure resembles the starting configuration.
Th–Cl bond distances in the relaxed 1D chain deviate maximum by 0.1 Å. The bond angles slightly deviate from the values calculated for the bulk ThCl₄. The deviation in the structural parameters can be attributed to the discontinuation of Th–Cl bonds present in the 1D chain in contrast to those observed in the bulk ThCl₄.

As discussed earlier, bulk ThCl₄ is a wide-bandgap semiconductor. The DOS of an infinite ThCl₄ chain outside the tube is slightly distorted compared to that of bulk ThCl₄. In particular, the alpha/beta (spin up/spin down) states are different. This is because some Cl atoms exhibit single coordination with adjacent Th atoms in contrast to what observed in bulk ThCl₄ [see Fig. 2(a)]. The bandgap for the 1D system (3.80 eV) is slightly larger than that calculated for the bulk (3.60 eV).

C. Calculations on ThCl₄@SWNT

Finally, we considered the modeling of 1D ThCl₄ encapsulated within SWNTs. Three different sizes of SWNTs were selected. They are (15,0), (16,0), and (17,0) with diameters of 11.85, 12.54, and 13.53 Å, respectively. The selection of the (15,0) tube is due to its diameter close to the value (11.00 Å) reported in the experiment. Additionally, we also considered two other tubes with larger diameters of 13.53 Å to look at the confinement effect of the diameter. A supercell comprising four ThCl₄ units (supercell length = 16.97 Å) and an (n,0) SWNT (n = 15, 16, and 17) comprising four unit cells (supercell length = 17.01 Å) were very close to each other. The number of carbon atoms in the pristine (15,0), (16,0), and (17,0) SWNTs is 240, 256, and 268, respectively. We applied periodic boundary conditions
TABLE III. Calculated encapsulation energies, bond lengths, bond angles, and charge transferred between the SWNT and the $\text{ThCl}_6$ chain.

| $d_t$ (Å) | $E_{\text{enc}}$ (eV) | $\text{Th}–\text{Cl}$ (Å) | $\text{Th}–\text{Cl}–\text{Th}$ (deg) | $\text{Cl}–\text{Th}–\text{Cl}$ (deg) | $\text{C}–\text{Cl}$ (deg) | Charge ($|\text{e}|$) |
|-----------|----------------|-----------------|----------------|----------------|----------------|---------------|
|           | $/\text{ThCl}_6$ unit |                 |                 |                 |                 | $/\text{ThCl}_6$ unit | |
|           |                 |                 |                 |                 |                 |               | |
| Dispersion |                 |                 |                 |                 |                 |               | |
| $\text{ThCl}_6@(15,0)$ | 11.85 | $-3.01$ | 2.71, 2.79/2.85, 2.87 | 107.29, 107.35 | 57.08, 71.09, 72.74, 92.69/141.73, 145.62 | 3.25, 3.27, 3.30, 3.48 | $-0.93$ |
| $\text{ThCl}_6@(16,0)$ | 12.54 | $-3.06$ | 2.72, 2.78/2.82, 2.89 | 107.30, 107.36 | 58.09, 68.32, 70.34/142.67, 145.78 | 3.26, 3.34, 3.45, 3.50, 3.47 | $-0.95$ |
| $\text{ThCl}_6@(17,0)$ | 13.53 | $-3.13$ | 2.71, 2.79/2.85, 2.90 | 107.29, 107.35 | 56.66, 68.98, 82.27, 90.5/143.26, 145.59 | 3.30, 3.37, 3.47 | $-0.92$ |
| Non-dispersion |                 |                 |                 |                 |                 |               | |
| $\text{ThCl}_6@(15,0)$ | 11.85 | $-1.54$ | 2.74, 2.77/2.84, 2.90 | 106.24, 106.39 | 60.03, 73.14, 75.46, 80.98/130.98, 143.45 | 3.33, 3.45, 3.56 | $-0.89$ |
| $\text{ThCl}_6@(16,0)$ | 12.54 | $-1.89$ | 2.76, 2.78/2.83, 2.87 | 107.34, 109.59 | 61.03, 72.14, 74.46, 81.98/132.98, 145.45 | 3.36, 3.43, 3.46 | $-0.89$ |
| $\text{ThCl}_6@(17,0)$ | 13.53 | $-2.14$ | 2.75, 2.79/2.86, 2.93 | 108.23, 109.06 | 59.65, 60.34, 85.34, 91.23/146.56, 147.45 | 3.39, 3.45, 3.56 | $-0.90$ |

The values calculated for the bond distances and bond angles deviate slightly from the values calculated for the gas phase 1d infinite $\text{ThCl}_6$ chain (see Table III), indicating a small distortion in the crystal upon encapsulation. In all cases, encapsulation energies are negative, meaning that they are thermodynamically stable inside the tube. For the smaller tube, the encapsulation energy is less exothermic due to a small distortion. Encapsulation energies were calculated without dispersion correction to look at the contribution of short-range van der Waals forces. Calculations clearly show to supercells of $\text{Th}_3\text{Cl}_{24}@(n,0)$ in order to model the entire supercell (SWNT+ThCl$_6$) as an infinite system. Figure 3 shows the modeled ThCl$_6$ chain encapsulated within a SWNT.

The values calculated for the bond distances and bond angles deviate slightly from the values calculated for the gas phase 1d infinite ThCl$_6$ chain (see Table III), indicating a small distortion in the crystal upon encapsulation. In all cases, encapsulation energies are negative, meaning that they are thermodynamically stable inside the tube. For the smaller tube, the encapsulation energy is less exothermic due to a small distortion. Encapsulation energies were calculated without dispersion correction to look at the contribution of short-range van der Waals forces. Calculations clearly show

![Figure 4](https://example.com/figure4)
FIG. 5. DOS plots of (a) pristine (15,0) tube, (b) pristine (17,0), (c) ThCl$_6$@(15,0), and (d) ThCl$_6$@(17,0).

FIG. 6. (a) Relaxed structure of an infinite ThCl$_6$ chain inside an infinite (17,0) tube, (b) its cross-sectional view, (c) charge density plot showing the interaction between the tube and the chain, and (d) its cross-sectional view.
that dispersion correction improves the encapsulation energies. The bond distances, bond angles, and the amount of charges are not significantly affected by the dispersion correction. Figure 4 shows the relaxed structure of ThCl$_6$@(15,0) and charge density of the composite. The distortion is further confirmed by the charge density plot in Fig. 4(d). The (15,0) tube is metallic [see Fig. 5(a)]. Upon encapsulation, the tube becomes a narrow gap semiconductor. This can be due to the charge transfer from the tube to the chain. The semiconducting nature (17,0) tube is still kept upon encapsulation though there is a reduction in the bandgap upon encapsulation. The inclusion of dispersion correction improved the calculated encapsulation energies. The calculated C–Cl bond distances (3.25–3.48 Å) confirm that there is a non-covalent interaction between the tube and the chain. This is further confirmed by the charge density plots (see Figs. 4 and 5).

IV. CONCLUSIONS

In conclusion, the encapsulation of the one-dimensional ThCl$_6$ chain within different sizes of SWNTs was studied using density functional theory with dispersion correction. The calculated structures are in good agreement with the structure observed in the HRTEM experiment. The encapsulation energies are exoergic, meaning that the chain structure inside SWNTs is thermodynamically stable. This is evidenced by the charge transfer between SWNTs and the chain. The inclusion of dispersion correction improved the encapsulation. There is a small reduction in the bandgap in the ThCl$_6$ chain compared to its bulk structure. Encapsulation introduces a gap in the metallic (15,0) tube, whereas the semiconducting nature of the (17,0) tube is retained though there is a reduction in the bandgap upon encapsulation.

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DATA AVAILABILITY

The data that support the findings of this study are available from the corresponding author upon reasonable request.

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